

Supporting Information for
Magnetic-Field Controllable Displacement-Type Ferroelectricity
Driven by Off-Center Fe²⁺ Ions in CaFe₃Ti₄O₁₂ Perovskite

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Table S1. Refined structure parameters of CaFe₃Ti₄O₁₂ based on the SXRD data collected at 290 K.

a (Å)	7.470873(2)
O_y	0.2984(1)
O_z	0.1887(1)
U_{iso} (Ca) ($100 \times \text{Å}^2$)	1.40(4)
U_{iso} (Fe) ($100 \times \text{Å}^2$)	2.54(2)
U_{iso} (Ti) ($100 \times \text{Å}^2$)	1.09(1)
U_{iso} (O) ($100 \times \text{Å}^2$)	0.94(1)
Ca-O ($\times 12$) (Å)	2.638(1)
Fe-O ($\times 4$) (Å)	2.0627(6)
	2.7707(6)
	3.2217(6)
Ti-O ($\times 6$) (Å)	1.9568(2)
Ti-O-Ti (°)	145.35(4)
Fe-O-Ti (°)	107.12(2)
BVS(Ca)	1.96
BVS(Fe)	1.96
BVS(Ti)	4.09
R_{wp} (%)	3.23
R_{p} (%)	2.19

The space group is $Im\bar{3}$ (No. 204), with atomic positions: Ca $2a(0, 0, 0)$, Fe $6b(0, 0.5, 0.5)$, Ti $8c(0.25, 0.25, 0.25)$, and O $24g(x, y, 0)$. The BVS values are calculated by using the formula $\sum_i \exp[(r_0 - r_i)/0.37]$, where r_i is the bond length, and r_0 is a constant for a specific ion. Here r_0 is 1.967 for Ca, 1.734 for Fe and 1.815 for Ti.

Table S2. Comparison of refined satisfactory goodness-of-fit parameters based on the SXRD pattern collected at 5 K using different space groups for CaFe₃Ti₄O₁₂

Space group	χ^2	R_{wp} (%)	R_p (%)
<i>Im-3</i>	0.1256	4.54	2.87
<i>R3</i>	0.0670	4.09	2.46
<i>Imm2</i>	0.0779	4.06	2.56

Table S3. Refined structure parameters of CaFe₃Ti₄O₁₂ based on the SXRD data collected at 5 K.

Atom	site	x	y	z	U _{iso} (100×Å ²)
Ca	3 <i>a</i>	0	0	-0.0050(8)	0.87(2)
Fe	9 <i>b</i>	0.1556(3)	0.3378(3)	0.3312(4)	3.62(1)
Ti1	3 <i>a</i>	0	0	0.4951(7)	0.30(1)
Ti2	9 <i>b</i>	0.5008(2)	-0.0024(4)	0.4894(4)	2.13(1)
O1	9 <i>b</i>	0.2271(7)	0.2641(5)	0.0892(8)	0.93(5)
O2	9 <i>b</i>	0.1551(4)	0.1693(6)	0.6904(7)	1.22(7)
O3	9 <i>b</i>	0.1693(5)	0.0308(4)	0.3424(9)	0.11(4)
O4	9 <i>b</i>	0.1208(4)	0.4189(6)	0.6421(7)	4.72(9)

Space group: *R*3 (No. 146), $a = 10.55453(3)$ Å, $c = 6.46314(4)$ Å, $R_{wp} = 4.09\%$, $R_p = 2.46\%$.

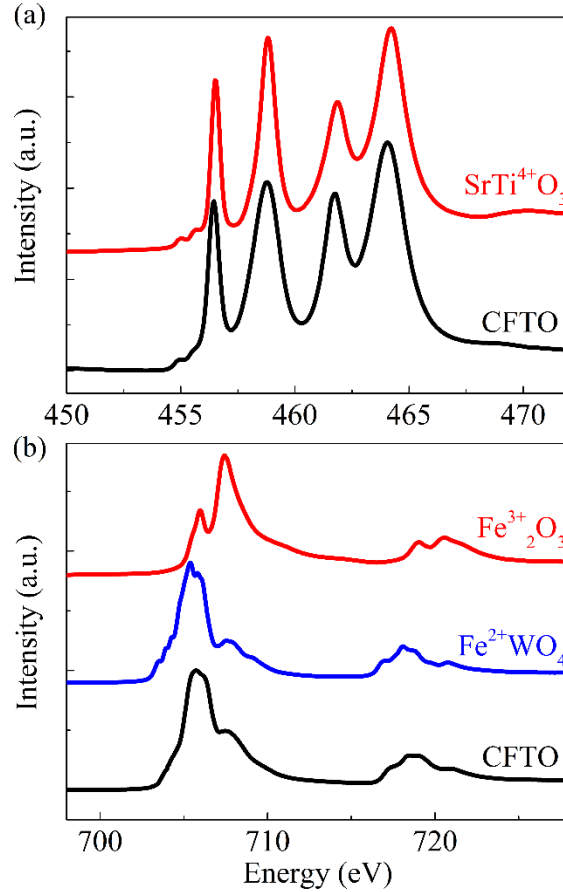


Figure S1. X-ray absorption spectroscopy for $\text{CaFe}_3\text{Ti}_4\text{O}_{12}$ (CFTO) measured at room temperature. (a) Ti- $L_{2,3}$ edges of CFTO and reference $\text{SrTi}^{4+}\text{O}_3$. Their similar spectral shapes and energy positions confirm the Ti^{4+} charge state in CFTO. (b) Fe- $L_{2,3}$ edges of CFTO and two references Fe_2O_3 and FeWO_4 [1] with Fe^{3+} and Fe^{2+} charge states, respectively. The similar energy positions and peak profiles between CFTO and FeWO_4 confirm the Fe^{2+} charge state in CFTO.

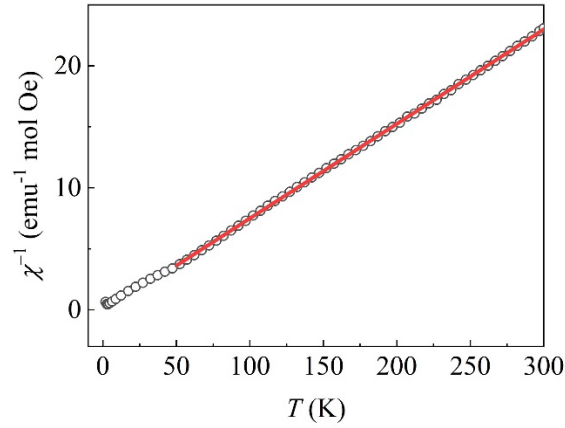


Figure S2. The Curie–Weiss fitting for the zero-field cooling (ZFC) data of χ^{-1} above 50 K using the function $\chi^{-1} = (T - \theta) / C$ for $\text{CaFe}_3\text{Ti}_4\text{O}_{12}$. The absolute value of the fitted Weiss temperature θ ($= -3.19$ K) is comparable to $T_N \approx 3.1$ K, and the negative sign agrees with the antiferromagnetic interaction. According to the fitted Curie Constant C ($= 12.91$ emu K Oe $^{-1}$ mol $^{-1}$), the effective magnetic moment is calculated as $10.13 \mu_B$ f.u. $^{-1}$. This value is somewhat larger than the theoretical one ($8.57 \mu_B$ f.u. $^{-1}$), if we only consider the contribution of the spin moment for a high-spin Fe^{2+} . The possible reasons are the presence of some magnetic moments at the oxygen sites owing to the presence of strong Fe–O covalent effects [2] and existence of the orbital contribution as proposed to explain the higher-than-expected effective Co^{2+} moment in isostructural $\text{CaCo}_3\text{Ti}_4\text{O}_{12}$ [3].

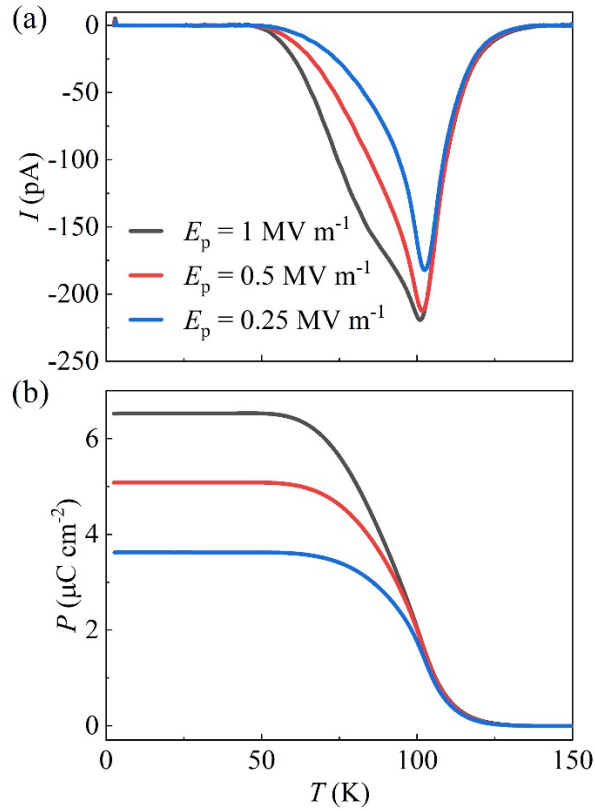


Figure S3. Temperature dependent (a) pyroelectric current and (b) corresponding polarization measured under different poling fields E_p . The pyroelectric current and the polarization increase with increasing poling fields, implying the polarization is not saturated, consistent with the PE loop results.

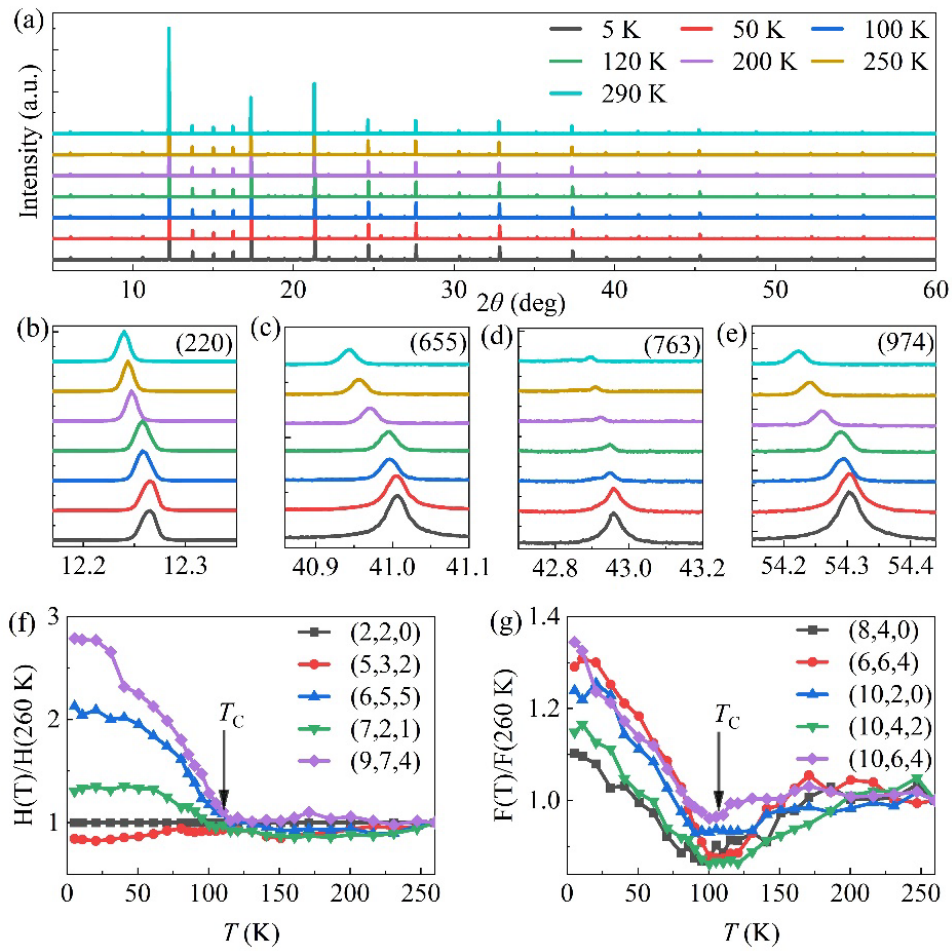


Figure S4. (a) SXR D patterns measured at some representative temperatures for $\text{CaFe}_3\text{Ti}_4\text{O}_{12}$. (b-e) Enlarge view of SXR D patterns for some characteristic diffraction peaks. Temperature dependence of, (f) peak height $H(T)$, (g) full width at half maximum (FWHM) $F(T)$ of some characteristic diffraction peaks.

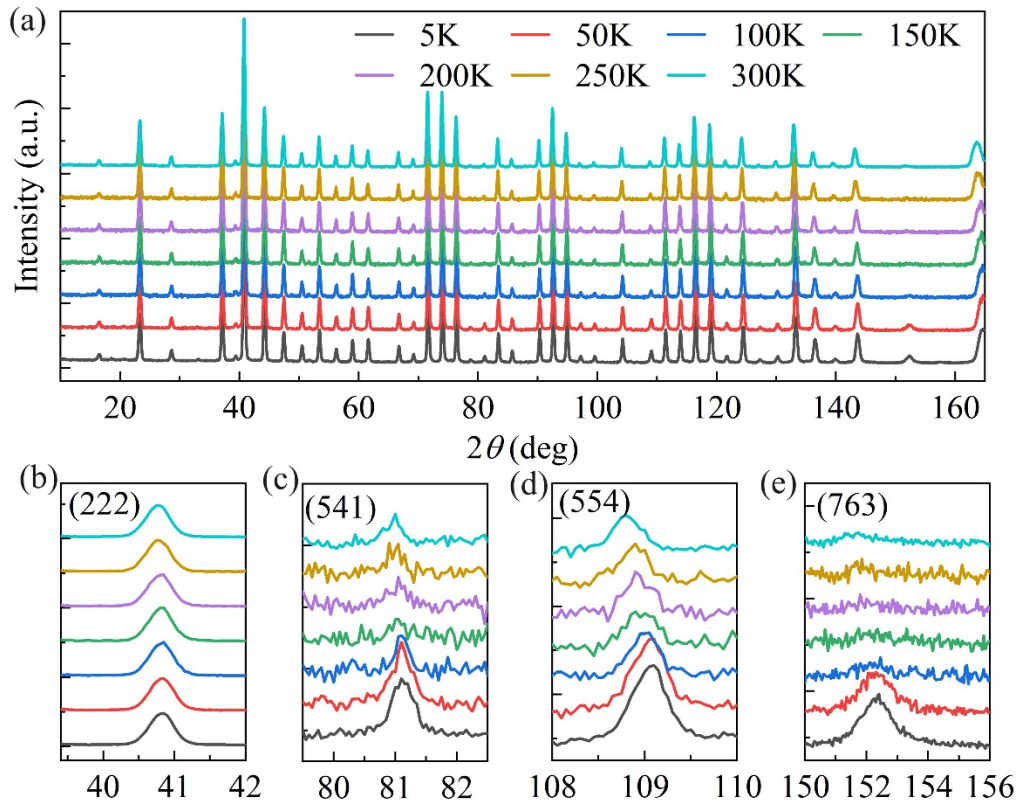


Figure S5. (a) NPD patterns measured at some representative temperatures with a wavelength of 1.494 \AA for $\text{CaFe}_3\text{Ti}_4\text{O}_{12}$. (b-e) Enlarge view of NPD patterns for some characteristic diffraction peaks. The relative intensities of some diffraction peaks increase remarkably below T_C , implying a structural transition.

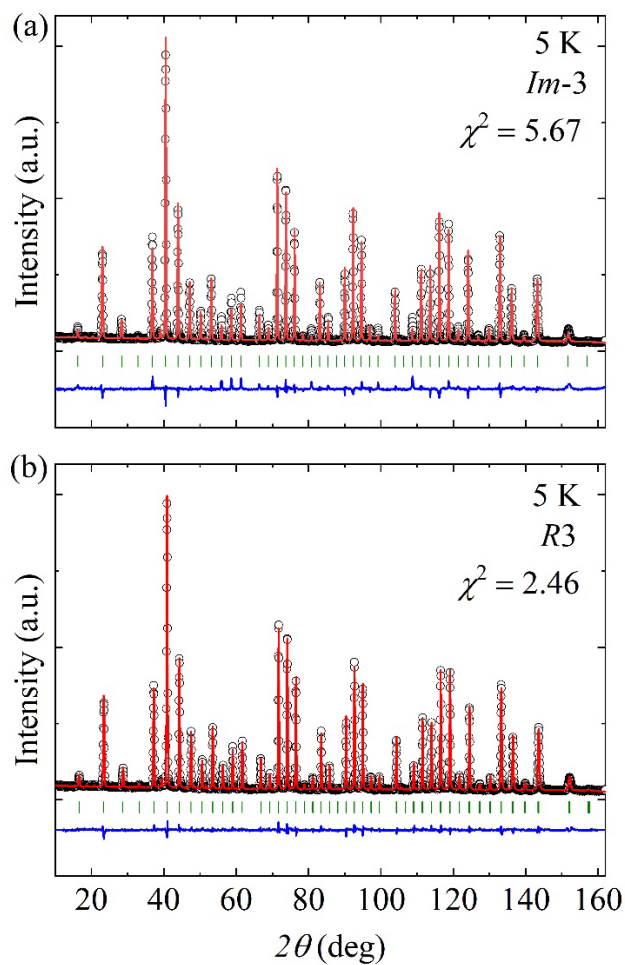


Figure S6. Rietveld refinement results based on the NPD data collected at 5 K using the (a) *Im-3* and (b) *R3* space group. Observed (black circles), calculated (red line), and difference (blue line) are illustrated. The green ticks indicate the allowed Bragg reflections. Compared with the refinement quality by using *Im-3* space group, significant improvement can be achieved by using *R3* space group.

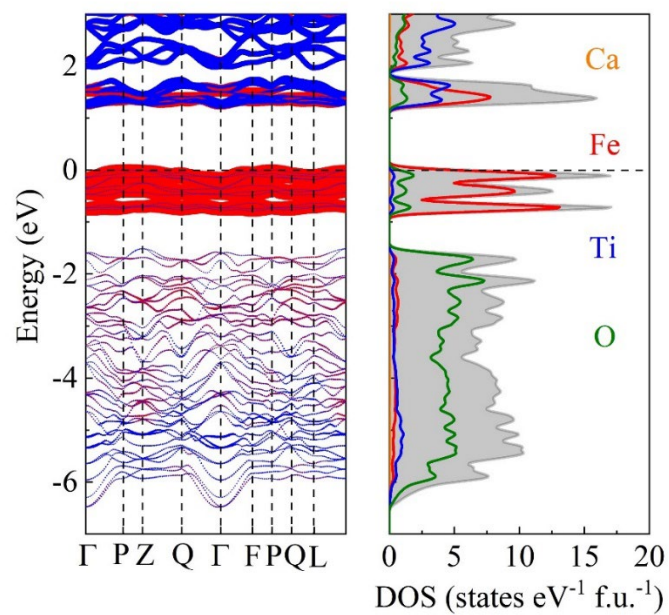


Figure S7. First-principles calculation results for the band structures and density of states (DOS) for $\text{CaFe}_3\text{Ti}_4\text{O}_{12}$ with the polarized $R3$ symmetry. The total DOS (light gray) and partial DOS of Ca (orange curves), Fe (red curves), Ti (blue curves), and O (green curves) are all shown for comparison.

Reference

1. Maignan A., Schmidt M., Prots Y. *et al.* FeWO₄ Single Crystals: Structure, Oxidation States, and Magnetic and Transport Properties. *Chem. Mater.* 2022; 34; 789-97.
2. Li H., Lv S., Han L. *et al.* First-principles investigation of magnetic coupling mechanism in A-site-ordered perovskite CaFe₃Ti₄O₁₂. *Comp. Mater. Sci.* 2012; **53**: 329-32.
3. Amano P. M., Denis R. F., Koo H.-J. *et al.* Orthogonal antiferromagnetism to canted ferromagnetism in CaCo₃Ti₄O₁₂ quadruple perovskite driven by underlying kagome lattices. *Commun. Mater.* 2022; **3**: 51.